

POSTER

HIGH-TEMPERATURE BONDING OF PARTS MADE OF CARBON AND CARBON-CERAMIC MATERIALS

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INTRODUCTION

A great number of methods have been developed to date for bonding articles made of carbon and carbon-ceramic materials (C-CM), which may be classified in terms of the distinctions of forming a bonding seam and physico-chemical processes occurring meanwhile. Methods of bonding C-CM may be conventionally divided into 3 main classes characterizing under which conditions the joint is formed: solid, liquid or vapor phases [1]. Wide application for bonding C-CM was found by a high-temperature, exothermal, and diffusion welding, as well as CVD methods [2]. For the high-temperature (HT) soldering of CM there have been developed solders based on Ti, Zr, Nb-alloys making it possible to bond CM between themselves or with refractory metals (W, Mo, Nb) at temperatures of 1500 to 2450 °C in vacuum or controlled media using a furnace or inductive heating. The exothermal soldering allows essential reduction of power consumption owing to heat emission in the exothermal reaction and secures short time at soldering temperature under reducing conditions typical of an exothermal reaction. The most promising HT adhesives (HTA) for joining CM are glue compositions based on polymeric carbonizable binders [3]. The main criteria when developing the adhesive formulations, were found to be a compatibility with the materials to be joined, a high resistance to thermal and chemical destructions in wide temperature range. It is found that such requirements are met by glue compositions based on phenolformaldehyde (PFR) and furan resins and powder fillers forming refractory compounds during the carbonization of the glue base. Depending on the filler material it is possible to form a C-composition glue seam or one based on

high-melting chemical compounds: mixtures of carbides with borides or silicides as well as mixed composition, such as C-carbide, C-carbide-boride, C-carbide-silicide

EXPERIMENTAL

As a subjects of investigations were taken HTA based on PFR differing in the powder filler composition as follows: K grade (Si), VK (B, Si), SVK (B, Si, ZrB₂), and UVK (B, Si, ZrB₂, C) which ensure a long-term service life of HTA joints in the temperature range of 1400 to 2000 °C and short-term at 2000-2800 °C in vacuum or inert atmosphere. In oxidizing media adhesive joints are serviceable up to the temperatures from 1600 to 1700 °C for a limit period of time [4].

Subject to investigation were: behavior of adhesively bonded samples of graphite and steel as well as siliconized graphite at different temperatures, and in 20% NaOH and H₂SO₄ solution at 20 and 90 °C for 30 days; behavior of HTA compositions in distilled and sea water for 90 days. As the performance criterion were taken changes in the adhesively bonded sample strength and in HTA composition mass during exposure. The results are presented in Tables 1, 2. Adhesive joints based on the HTA K are highly resistant to distilled and sea water, to 20% H₂SO₄ both at 20 and 90 °C (Tables 1 and 2). To bond CM working at temperatures between 1400 and 1700 °C HTA VK grade have been developed. Its composition hardens at temperature of 130±20 °C for 3-4 h. The spreadable life of the HTA K is up to 5 days. Due to boron presence HTA VK is superior to the K adhesive in refractoriness in the temperature range from 800 to 1600 °C as well as in strength in the low temperature range. The most universal are HTA of the SVK and UVK grades. Due to introduct-

ion high-melting borides into their filler, a number of technical properties of these HTA exceed those of the HTA VK the limiting working temperature which is 1900 oC. The SVK and UVK adhesives ensure a long-term serviceability of the bonded structures made of CM up to the temperature of 1800 to 1900 oC in inert gases or under vacuum of 0.13-0.01 Pa, and a short-term serviceability at temperatures up to 2700 oC. The UVK adhesive composition is distinct from the SVK one in that the filler contains C components in the form of PFR coke having different HTT which is cheaper without marked decrease in its strength characteristics as indicated below:

Temperature, oC	Ultimate strength, MPa, at tear-off	shear
	V-1 graphite + V-1 graphite	
20	10 to 15	10 to 12
1800	9 to 12	8 to 10
2000	3 to 4	2.5 to 3.0
2300	1.2 to 2.0	1.5 to 2.0
	V-1 + 1Cr17Ni14Mo2Ti steel	
20	8 to 12	5.5 to 7.5
600	2 to 3	-
800	0.5 to 1.0	-

The HTA SVK and UVK are highly resistant in distilled and sea water; their corrosion resistance markedly exceeds that of the K and VK adhesives.

Table 1. Mass loss (Δm) of the adhesive compositions in H₂SO₄ (*) and NaOH (+) 20 % solutions

Adhesive composition	Δm , %, at 20 oC and exposure for (day)			Δm , %, at 90 oC		
	5	15	30	5	15	30
K	0.3	0.7	1.4	n/ch	1.2	1.5*
	63.5	65.0	67.5	69.5	74.5	80.0+
VK	1.0	1.35	1.8	3.4	5.5	13.0*
	39.8	40.5	42.0	42.0	45.5	56.0+
SVK	0.65	1.05	1.65	n/ch	n/ch	2.8*
	16.0	17.5	19.0	18.5	21.0	27.5+
UVK	0.3	0.6	1.0	n/ch	0.7	1.0*
	23.0	23.5	24.0	25.5	31.5	43.5+

CONCLUSIONS

Oxidation resistance of the HTA compositions in air increases with the content increase of the components ensuring the formation of borosilicate protective films of the ZrO₂-SiO₂-B₂O₃ system upon reacting with O₂. The corrosion resistance of the cured HTA compositions and the adhesive joints in NaOH increases with the increase of carbide phases and decrease of free Si content therein but in an acidic medium it is markedly higher than in an alkaline one and more essentially depends on the medium temperature. The resistance of the HTA compositions and the adhesively bonded joints on their basis is markedly higher in distilled water than in sea water.

REFERENCES

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Table 2 Resistance of the adhesive materials and joints in distilled and sea water after exposure for 90 days

Adhesive composition	Mass loss, %		Tear-off strength loss, %	
	Distil. water	Sea water	Distil. water	Sea water
K	1.15	1.25	8.5	36.0
VK	1.35	1.20	8.2	28.0
SVK	1.10	0.75	1.65	24.0
UVK	0.80	0.80	1.65	25.0